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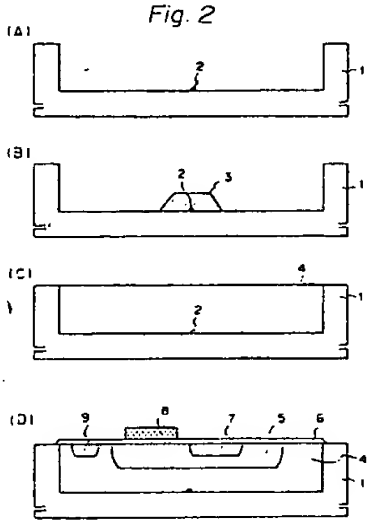
(54) **Photoelectric converter.**

(57) A photoelectric converter comprises a semiconductor transistor having a main electrode region of one conductivity type semiconductor and a control electrode region of an opposite conductivity type semiconductor, and a capacitor for controlling a potential of the floating control electorde region. Carriers generated by a light are stored in the control electrode region by controlling the potential of the control electrode area through the capacitor. The stored voltage is read and stored carriers are refreshed.

On an unnucleation surface (S_{NDS}) of said light transmissive substrate, a nucleation surface (S_{NDL}) of a heterogeneous material having a sufficiently higher nucleation density than that of the material of unnucleation surface and a sufficiently small area to permit growth of single nuclear of the deposite surface is formed. A single crystal region is grown around the single nuclear formed in the nucleation surface (S_{NDL}). And a photoelectric conversion cell is formed is the single crystal region.

Fig. 2



EP 0 260 955 A2

Description

Photoelectric Converter

BACKGROUND OF THE INVENTION

Field of the Invention

The present invention relates to a semiconductor device such as a photoelectric converter having a photoelectric charge storage area having a potential thereof controlled through a capacitor.

Related Background Art

A photoelectric converter having a photoelectric charge storage area having a potential thereof controlled through a capacitor has been known, for example, by European Patent Publication No. 132076.

It is briefly explained below.

Fig. 1(A) shows a plan view of the photoelectric converter and Fig. 1(B) shows a I-I sectional view thereof.

Photoelectric conversion cells are arranged on an n-type silicon substrate 101 and each photoelectric conversion cell is electrically insulated from adjacent photoelectric conversion cells by a device isolation area 102 made of SiO_2, Si_3N_4 or polysilicon.

Each photoelectric conversion cell is constructed as follows. A p region 104 is formed on a low impurity density n- region 103 formed by epitaxial technique by doping p-type impurity (for example, boron), and an n+ region 105 is formed in the p region 104 by impurity diffusion technique or ion injection technique. The p region 104 and the n+ region 105 are base and emitter of an NPN bipolar transistor, respectively.

An oxidation film 106 is formed on the n- region 103 having the respective regions formed thereon, and a capacitor electrode 107 having a predetermined area is formed on the oxidation film 106. The capacitor electrode 107 faces the p region 104 with the oxidation film 106 being interleaved therebetween, and a potential of the floating p region 104 is controlled by applying a pulse voltage to the capacitor electrode 107.

An emitter electrode 108 connected to the n+ region 105, a wiring 109 for externally reading a signal from the emitter electrode 108, a wiring 110 connected to the capacitor electrode 107, a high impurity concentration n+ region 111 on the rear side of the substrate 101, and an electrode 112 for applying a potential to the collector of the bipolar transistor are formed.

A basic operation is now explained. It is assumed that the p region 104 which is the base of the bipolar transistor is initially at a negative potential. A light is applied to the p region 104 and holes of electron-hole pairs generated by the incident light are stored in the p region 104 so that the potential of the p region 104 rise toward a positive level (store operation).

Then, a read positive voltage pulse is applied to the capacitor electrode 107 while the emitter electrode 108 floats. When the positive voltage is applied to the capacitor electrode 107, the potential of the p region 104 which is the base rises so that the base-emitter is forward biased and a current flows between the collector and the emitter in accordance with the increment of the base potential at the storage operation. Accordingly, an electrical signal corresponding to the incident light intensity appears at the floating emitter electrode 108 (read operation). Since the charge stored in the p region 104 (base) does not substantially decrease, the same optical information can be repetitively read.

In order to remove the holes stored in the p region 104, the emitter electrode 108 is grounded and a refresh positive voltage pulse is applied to the capacitor electrode 107 so that the p region 104 is forward biased relative to the n+ region 105 and the stored holes are removed through the grounded emitter electrode 108. When the refresh positive voltage pulse falls, the base potential of the p region 104 resets to the initial negative potential (refresh operation). Thereafter the store, read and refresh operations are repeated.

In the proposed system, the charge generated by the light irradiation is stored in the base p region 104 and the current flowing between the emitter electrode 108 and the collector electrode 112 is controlled by the stored charge. Accordingly, the stored charge is read after amplification by the amplification functions of the cells and hence high output, high sensitivity and low noise are attained.

However, this photoelectric converter has a problem described below.

When areas of highly density integrated photoelectric conversion cells are very small, an aperture factor is very low and a wiring capacitance increases. In other words, a potential V_p generated by the holes stored in the base by the light excitation decreases.

The increase of the wiring capacitance may be prevented by increasing a thickness of the insulative layer between the photoelectric conversion cell and the wiring material or using a low dielectric constant material, but the reduction of the aperture factor cannot be essentially prevented because the wiring electrode and the device isolation area exist on the photoelectric conversion cell. As a result, the larger the circuit scale is and the higher the integration density is, the lower is the output level.

In the prior art method, the die or pigment used for an on-chip color filter is deposited directly on the cell. As a result, contamination materials in the die or pigment easily go into the cell. This cause a problem in reliability.

SUMMARY OF THE INVENTION

It is an object of the present invention to provide a photoelectric converter comprising a semiconductor transistor having a main electrode region of one conductivity type semiconductor and a control electrode region of the opposite conductivity type semiconductor, and a capacitor for controlling a

potential of the floating control electrode region to store carriers generated by a light in the control electrode region, read the stored voltage and refresh the stored carriers, characterized by that deposition surface (S_{NDL}) of a heterogeneous material which has a sufficiently higher nuclear formation density than that of the material forming the deposition surface (S_{NDS}) and of which area is fine enough to permit growth of only one nuclear of the deposit material is formed on the deposit of a light transmissive substrate, and photoelectric cells are formed in a single crystal region grown around the single nuclear formed in the heterogeneous material.

In accordance with the present invention, the single crystal region is formed in the deposit on the recess formed in the light transmissive substrate, and the photoelectric conversion cells are formed in the single crystal region so that the light may be applied to a back side.

The single crystal region may be formed by a semiconductor process described as follows.

There is no restriction on a substrate material such as that in the SOS (silicon-on-saphire) technique, and no diffusion of aluminum to the cell occurs.

Accordingly, the aperture factor is improved, high integration density is attained and the reliability of the cell is improved because comtamination material does not go into the cell.

BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1(A) shows a plan view of a photoelectric converter,

Fig. 1(B) shows a I-I sectional view thereof,

Figs. 2(A) - 2(E) show longitudinal sectional views for illustrating a manufacturing step of one embodiment of the photoelectric converter of the present invention,

Figs. 3(A) and 3(B) show longitudinal sectional views for illustrating a manufacturing process of a photoelectric converter having an improved characteristic,

Fig. 4 shows a longitudinal sectional view of a second embodiment of the photoelectric converter of the present invention,

Fig. 5 shows a longitudinal sectional view of a third embodiment of the photoelectric converter of the present invention,

Fig. 6 shows a longitudinal sectional view for illustrating a manufacturing process of a fourth embodiment of the photoelectric converter of the present invention,

Figs. 7(A) and 7(B) illustrate a selective deposition method,

Fig. 8 shows a graph of nuclear formation densities on a $S:O_2$ deposit and a silicon nitride deposit,

Figs. 9(A) - 9(D) show steps of a single crystal formation method,

Figs. 10(A) and 10(B) show perspective views of a substrate in Figs. 7(A) - 7(D),

Fig. 11 shows a graph of flow rates of SiH_4 and NH_3 and a composition of Si and N of the silicon nitride film,

Fig. 12 shows a graph of the Si/N composition and the nuclear formation density, and

Fig. 13 shows a graph of a Si ion injection rate and the nuclear formation density.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

Figs. 2(A) - 2(E) show longitudinal sectional views for illustrating a manufacturing process of a first embodiment of the photoelectric converter of the present invention.

As shown in Fig. 2(A), a recess of necessary size and shape is formed in a light transmissive substrate 1, a heterogeneous material having a high nuclear formation density different from concave structure material of light transmissive substrate is thinly deposit thereon and it is patterned by lithography to form a very fine nucleation deposition surface (S_{NDL}) formed of heterogeneous material 2. The heterogeneous material 2 may be formed of a modified region having excess Si or Ni formed by ion-injection of Si or Ni into the thin film.

Then, a single nuclear of the thin film material is formed only in the heterogeneous deposition surface (S_{NDL}) 2. The heterogeneous deposition surface (S_{NDL}) 2 must be fine enough to permit the formation of only one nuclear. The size of the heterogeneous deposition surface (S_{NDL}) 2 may be less than several microns, although it may vary with the material. As shown in Fig. 2(B), the nuclear is grown while it maintains the single crystal structure to form an island-shaped single crystal grain 3. The island single crystal grain 3 is further grown around the heterogeneous deposition surface (S_{NDL}) 2 while it maintains the single crystal structure, until the single crystal grain 3 fills out the recess. Thus, a single crystal region 4 shown in Fig. 2(C) is formed. A portion of the region 4 serves as a collector region 4-A of resulted photoelectric conversion device.

As shown in Fig. 2(D), a base region 5 is formed in a portion of the collector region 4 by using ion implant method or thermal diffusion method. Then, a gate oxidation film 6 is formed on a surface of the single crystal region 4 by dry oxidation method or wet oxidation method by pyrogenic. A polysilicon electrode 8 is formed on a portion of the base region 5 by a LPCVD method. The polysilicon electrode 8 may be a high melting point metal such as W or Mo or siliside or polyside thereof. When the polysilicon is used by the LPCVD method, a film is formed at a temperature of 600 - 650°C, a pressure of 0.2 - 1.0 Torr and a SiH_4 flow rate of 20 - 100 sccm, and P is highly doped to a visinity of solid solution limit at 950°C by deposition by $POC\ell_3$ or Ion implantation. Then, ions are implanted to an emitter region 7 and a collector contact region 9 by using a mask pattern formed by photoresist.

As shown in Fig. 2(E), an interlayer insulation layer 10 is deposited by CVD method or bias sputter method. Contact holes are formed by photo-etching, and electrodes 11 and 12 of the emitter region 7 and the collector contact region 9 are formed by $A\ell$, $A\ell$-Si, W, Mo, W siliside, Mo siliside, Ti or Ti siliside. In the present embodiment, since the substrate is made of a light transmissive material, the light may

be irradiated to a back side o ubstrate opposite to the recess. As a result, in the present embodiment, the light is not blocked by the electrodes and Aℓ wiring which were the causes of light blocking in the prior art converter. Accordingly, the characteristic of the photo-conversion cell can be improved without paying attention to the aperture factor of the cell. Figs. 3(A) and 3(B) show longitudinal sectional views for illustrating a manufacturing process of the photoelectric converter having the improved characteristic.

The process is identical to that of Fig. 2 until the gate oxidation film 6 is formed in Fig. 2(D). To compare with Fig. 2(E), a refresh PMOS transistor is added to the base region. It may be formed in the following manner.

After the gate oxidation film 6 has been formed, the base capacitor electrode 8 and the PMOS transistor gate electrode 13 are simultaneously formed by photo-etching by using polysilicon, or Mo or W as shown in Fig. 3(A). The collector contact region 9 and the emitter region 7 are then formed by the ion implantation, and a PMOS source region 14 and a drain region 15 are formed by the ion implantation by using a self-alignment method.

As shown in Fig. 3(B), the interlayer insulation film 10 is deposited by the CVD method or the bias sputter method. Then, the collector electrode 12, emitter electrode 11 and PMOS source electrode 16 are formed. The PMOS is turned on in the refresh operation. As a result, the base potential is kept constant irrespective of the stored charge and high speed refresh operation is attained and a residual image is prevented.

Since the light is irradiated to the back side of the converter, the formation of the PMOS on the front surface of the converter does not affect to the aperture factor.

A second embodiment of the photoelectric converter of the present invention is now explained.

Fig. 4 shows a longitudinal sectional view of the second embodiment. It is manufactured in the same process as that of the first embodiment until the step of Fig. 2(E). In the present embodiment, a V-shaped groove 17 is formed by photo-etching in the back side of the light transmissive substrate 1. Preferably, the etching is done as isotropically as possible, and either dry etching or wet etching may be used. In accordance with the present embodiment, the V-shaped groove may be formed in the isolation region between the adjacent photoelectric conversion cells so that a light irradiated to the isolation region is refracted to the single crystal region to effectively irradiate it. Further, a cross-talk is reduced.

The back side process may be done after the photoelectric conversion cells have been formed or may be done prior to the formation of the photoelectric conversion cells and then the single crystal region and the photoelectric conversion cells may be formed.

In the present invention, the back side process for efficiently collecting the light is not limited to the V-shaped groove but it may be done by a lens having a curved surface.

A third embodiment of the photoelectric converter of the present invention is now explained.

Fig. 5 shows a longitudinal sectional view of the third embodiment. It is manufactured by the same process as that of the first embodiment until the step of Fig. 2(E).

In the present embodiment, the light transmissive substrate 1 has a light transmissive layer 18 of a different optical property added thereto, the layer 18 may be a non-reflection layer for suppressing diffused reflection of the incident light on the substrate surface, or a color filter layer.

The non-reflection layer may be a single layer (MgF_2) of MgF_2, ZrO_2, and $A\ell_2O_3$ or laminated plural layers of them.

The color filter layer may be formed by appropriately combining yellow, cyan and magenta dies or blue, red and green filter films by using photo-etching or lift-off technique. The filter film may be formed by a single layer of perylene pigment or a laminated layer of perylene pigment and isoindolinone pigment for a red filter, a single layer of starocyanine pigment or a laminated layer of starocyanine pigment and isoindolynone pigment for a green filter, and a single layer of starocyanine for a blue filter.

A fourth embodiment of the photoelectric converter of the present invention is explained.

Fig. 6 shows a longitudinal sectional view of a manufacturing step of the fourth embodiment.

The photoelectric converter of the present embodiment produces a reference output in a dark state and is used with a photo-sensing photoelectric conversion cell.

A recess is formed in the light transmissive substrate 1, and a polysilicon film 19 is deposited by the LPCVD method. The polysilicon film 19 is formed to block the light. It is removed except for the cell area.

The polysilicon is then thermally oxidized to form a SiO_2 layer 20, and a heterogeneous deposition surface (S_{NDL}) 2 for nuclearization is formed of material different from SiO_2. The subsequent steps are identical to those of Figs. 2(B) - 2(E).

In the present embodiment, since the entire single crystal region is covered by the polysilicon film 19 and the light is blocked, the affect by the light irradiation is substantially eliminated.

The output of such a dark cell corresponds to a dark signal of other photoelectric conversion cell. Accordingly, by subtracting the output of the dark cell from the output of the other photoelectric conversion cell, an electrical signal which exactly represents the incident light is produced.

A method for forming the single crystal region 4 on the light transmissive substrate 1 is explained in detail concretely.

In order to make complete the understanding of the present invention a selective deposition method for selectively forming a deposition film on the deposit is first explained. In the selective deposition method, the thin film is selectively formed on the substrate by utilizing a difference between factors of materials which are influential to the nuclearization in the thin film formation process such as surface energy, deposition coefficient, separation coefficient

and surface diffusion rate.

Figs. 7(A) and 7(B) illustrate the selective deposition method. As shown in Fig. 7(A), a thin film 22 of a material having the above factor different from that of the substrate 21 is formed on the substrate 21. If the thin film of an appropriate material is deposited under an appropriate deposition condition, the thin film 23 grows only on the thin film 22 and does not grow on the substrate 21. By utilizing the phenomenon, the thin film 23 formed in self-alignment can be grown and the resist or lithography process used in the prior art process is not necessary.

The materials for such selective deposition method are SiO_2 for the substrate 21, and Si, GaAs or silicon nitride for the thin film 22 and Si, W, GaAs and Inp for the thin film 23 to be deposited.

Fig. 8 shows a graph of nuclearization densities on the deposit of SiO_2 and the deposit of silicon nitride. As shown, the nuclearization density on SiO_2 saturates at less than 10^3 cm^{-2} soon after the start of deposition, and the value does not significantly change 20 minutes later.

On the other hand, the nuclearization density on the silicon nitride (Si_3N_4) saturates at 4×10^5 cm^{-2} and does not change for ten minutes, but it abruptly increases thereafter. In this example, the $SiC\ell_4$ gas is diluted by the H_2 gas and deposition is made at a pressure of 175 Torr and a temperature of 1000°C by the CVD method. The same action may be attained by using reaction gas of SiH_4, $SiH_2C\ell_2$, $SiHC\ell_3$ or SiF_4 and adjusting the pressure and temperature. Vacuum deposition may be used.

The nuclearization on the SiO_2 raises no substantial problem. By adding the $HC\ell$ gas to the reaction gas, the nuclearization on the SiO_2 can be further suppressed so that the deposition of Si on the SiO_2 is completely blocked.

Such phenomenon is largely due to the difference between the SiO_2 and the silicon nitride of absorption coefficient, separation coefficient and surface diffusion coefficient to Si but it is also the cause of the selective deposition that SiO_2 reacts by the Si atoms themselves, SiO_2 is etched by the generation of silicon monoxide having a high vapor pressure while the etching does not occur on the silicon nitride. (Journal of Applied Physics, 53, 6839, 1982, by T. Yonehara, S. Yoshioka and S. Miyazawa).

By selecting SiO_2 and silicon nitride as the materials for the deposit surface and silicon as the deposit material, a sufficiently large nuclearization density difference (Δ_{ND}) is attained as shown in Fig. 8. The material of the deposit surface is preferably SiO_2 but SiO_2 may be used to attain a large nuclearization density difference (Δ_{ND}).

Those materials are not restrictive but any other materials may be used so long as one nuclearization density difference (Δ_{ND}) is at least 10^3 times as high as that of the other. The materials described hereinafter can also be used for the selective deposition.

Another method to attain the nuclearization density difference (Δ_{ND}) is to locally implant Si or N ions to SiO_2 to form a region which includes excess Si or N. According to the present invention, by using the selective deposition method on the basis of the nucleation density difference (Δ_{ND}), and forming the deposition surface of the heterogeneous material having a sufficiently larger nuclearization density than that of the material of the deposit surface into a sufficiently fine are to permit the formation of only one nuclear, the single crystal can be selectively grown at the area where the fine heterogeneous material is present.

Since the selective growth of the single crystal is determined by the electron state of the deposit surface, particularly by a dungling bond state, a material having a low nuclearization density (for example, SiO_2) need not be bulk material and it may be formed on only material or substrate.

Figs. 9(A) - 9(D) show a manufacturing process of the single crystal according to the present invention. Figs. 10(A) and 10(B) show perspective views of substrate in Figs. 9(A) and 9(D).

As shown in Figs. 9(A) and 10(A), the thin film 25 having a low nuclearization density to allow selective deposition is formed on the substrate 24, the heterogeneous material having a high nuclearization density is thinly deposited thereon, and it is patterned by lithography to form the sufficiently fine deposition surface (S_{NDL}) 26. The size, crystal structure and composition of the substrate 24 may be arbitrary and the substrate may include a functional device of conventional semiconductor material. The deposition surface (S_{NDL}) 26 of heterogeneous material includes a modified area having excess Si or N formed by ion-implanting Si or N into the thin film 25.

The single nuclear of the thin film material is formed only on the deposition surface (S_{NDL}) 26 of the heterogeneous material under an appropriate deposition condition. The deposition surface (S_{NDL}) 26 of the heterogeneous material must be formed sufficiently finely to permit the formation of the single nuclear. The size of the deposition surface (S_{NDL}) 26 may be less than several microns although it may vary with the material. The nuclear grows while it maintains the single crystal structure into the island-like single crystal grain 27 as shown in Fig. 9(B). As described above, it is necessary to establish a condition such that the nuclearization does not take place on the thin film 25.

The island-like single crystal grain 27 further grows around the deposition surface (S_{NDL}) 26 while it maintains the single crystal structure and finally covers the entire thin film 25 as shown in Fig. 9(C).

Then, the single crystal grain 27 is flattened by etching or polishing so that the single crystal layer 28 in which a desired device may be formed is formed on the thin film 25 as shown in Figs. 9(D) and 10(B).

Since the thin film 25 is formed on the substrate 24, the material of the supporting substrate 24 may be arbitrary, and even if the substrate 24 has a functional device formed thereon, the singal crystal layer can be readily formed thereon.

In the above embodiments, the surface of substrate 24 comprises the thin film 25. Alternatively, the single crystal layer may be formed by using a substrate made of low nuclearization density material to allow selective deposition.

Example

A specific example of forming the single crystal layer is explained.

SiO_2 is used as a deposit surface forming material for the thin film 25. A quarts substrate may be used as the substrate, or an SiO_2 layer may be found on a substrate such as metal, semiconductor, magnetic material, piezoelectric material or insulative material by sputtering, CVD or vacuum deposition. The thin film 25 forming material is preferably SiO_2 although SiOx may also be used.

A silicon nitride (Si_3N_4) layer or polycrystalline silicon layer is deposited as the deposition surface (S_{NDL}) forming material on the SiO_2 layer 25 by a depressured vapor deposition method, and the silicon nitride layer or polycrystalline silicon layer is patterned by a conventional lithography technique or a lithography technique which uses X ray, electron beam or ion beam, to form the fine deposition surface (S_{NDL}) 26 of less than several microns, preferably less than 1 micron. Size of the surface (S_{NDL}) may be several microns preferably smaller than 1 micron.

Then Si is selectively grown on the substrate 11 by using gas mixture of $HC\ell$, H_2 and $SiH_2C\ell_2$, $SiC\ell_4$, $SiHC\ell_3$, SiF_4 or SiH_4. The substrate temperature is 700 - 1100°C and the pressure is approximately 100 Torr.

The single crystal Si grain 27 is grown around the fine deposition surface (S_{NDL}) 26 of heterogeneous material of the silicon nitride or polycrystalline silicon on the SiO_2 in several tens minutes. It grows to more than several tens microns in size under an optimum growth condition.

Then, only the Si is etched and flattened by reactive ion etching (RIE) having an etching rate difference between Si and SiO_2 so that the polycrystalline silicon layer having a controlled grain size is formed. A grain boundary is removed and the island-like single crystal silicon layer 28 is formed. When the surface roughness of the single crystal grain 27 is high, the surface is first mechanically ground and then etched.

A field effect transistor is formed in the single crystal silicon layer 28 of more than several tens microns in size and having no grain boundary. The field effect transistor exhibited a no inferior property to a field effect transistor formed on a single crystal silicon wafer.

Since the device is electrically isolated from the adjacent single crystal silicon layer 28 by SiO_2, there is no mutual interference even if complementary field effect transistors (C-MOS) are fabricated. Since an activation layer of the device is thinner than that when the Si wafer is used, a malfunction due to charge in the wafer created when a radiation beam is emitted is eliminated. Since a stray capacitance is reduced, an operation speed of the device is improved. Since any substrate material may be used, the single crystal layer may be formed on a large substrate with a lower cost than that required when the Si wafer is used. Since the single crystal layer may be formed even on other semiconductor, piezoelectric material or dielectric substrate, a multifunction three-dimensional integrated circuit can be attained.

Composition of Silicon Nitride

In order to attain a sufficient nuclearization density difference between the deposit surface (crystal growth surface) and the deposition surface (S_{NDL}) of material different from the former surface, the composition of silicon nitride is not limited to Si_3N_4 but other composition may be used.

In the plasma CVD method which forms the silicon nitride film at a low temperature by decomposing SiH_4 gas and NH_3 gas in an RF plasma, flow rates of SiH_4 gas and NH_3 gas are changed to control the composition of Si and N of the deposited silicon nitride.

Fig. 11 shows a relation between a flow rate ratio of SiH_4 and NH_3 and a composition of Si and N in the silicon nitride film.

The deposition condition is an RF output of 175 watts and a substrate temperature of 380°C. The SiH_4 gas flow rate is fixed to 300 cc/min and the NH_3 gas flow rate is changed. When the flow rate ratio of NH_3 and SiH_4 gases is changed between 4 and 10, the Si/N ratio in the silicon nitride changes between 1.1 and 0.58 as proved by an electron spectrography.

When $SiH_2C\ell_2$ gas and NH_3 gas are introduced by the depressured CVD method and the silicon nitride is formed at 0.3 Torr and 800°C, the composition of the silicon nitride is close to storichiometric composition of Si_3N_4 (Si/N = 0.75).

When the silicon nitride film is formed by heat treatment of Si at 1200°C in anmonium or N_2 gas (thermal nitrization), the composition is closer to the storichiometric composition because the formation is under thermal balance.

By growing the nuclear of Si by using the silicon nitride thus formed as the deposit surface having higher nuclearization density than that of SiO_2, the nuclearization density difference is attained in accordance with the composition.

Fig. 12 shows a relation between the Si/N composition and the nuclearization density. As shown, as the composition of the silicon nitride changes, the nuclearization density of Si grown thereon significantly changes.

The Si is grown at a $SiC\ell_4$ gas pressure of 175 Torr, and a H_2 reaction temperature of 1000°C.

The phenomenon of change of the nuclearization density by the composition of the silicon nitride influences to the size of the silicon nitride which is the heterogeneous material formed sufficiently finely to permit the growth of single nuclear. That is, the silicon nitride having a large nuclearization density cannot form single nuclear unless it is very finely formed.

Accordingly, it is necessary to select the nuclearization density and an optimum size of the silicon nitride to permit the growth of single nuclear. For example, under the deposition condition which attains the nuclearization density of approximately 10^5 cm^{-2}, a single nuclear may be formed if the size of the silicon nitride is less than 4 microns.

Formation of deposition surface (S_{NDL}) by ion implantation.

In order to attain the nuclearization density difference to Si, ions of Si, N, P, B, F, Ar, He, C, As, Ga or Ge may be locally implanted to the SiO_2 which is a material for forming the deposit surface having a low nuclearization density to form a modified area in the deposit surface of SiO_2, and the modified area may be used as a material for forming the deposit surface (S_{NDL}) having a high nuclearization density.

For example, the surface of SiO_2 is covered by a resist and desired areas are exposed to light, developed and dissolved to partially expose the SiO_2 surface.

Then, the SiF_4 gas is used as a source gas and Si ions are implanted to the SiO_2 surface at 10 KeV at a density of $1 \times 10^{16} - 1 \times 10^{18}$ cm^{-2}. A projection fly path is 114 Å and the Si density on the SiO_2 surface is approximately 10^{22} cm^{-3}. Since SiO_2 is inherently amorphous, the Si ion-implanted area is also amorphous.

In order to form the modified area, ions may be implanted by using the resist as a mask but the focused Si ions may be implanted to the SiO_2 surface by focused ion beam technique without using the resist mask.

After the ion implantation, the resist is removed and the modified area including excess Si is formed on the SiO_2 surface. Then, Si is vapor grown on the SiO_2 deposit surface having the modified area formed therein.

Fig. 13 shows a relation of Si ion implantation quantity and the nuclearization density. As shown, the larger the Si+ implantation quantity, the more does the nuclearization density increase.

Accordingly, if the modified area is formed sufficiently finely, single Si nuclear can be grown by using the modified area as the heterogeneous material and the single crystal can be grown as described before.

The formation of the modified area in a sufficiently fine area to permit the growth of single nuclear may be readily attained by patterning the resist or focusing the ion beam.

Method for Depositing Si Other Than CVD

In order to grow the single crystal by the selective nuclearization of Si, a method for evaporating Si in vacuum ($< 10^{-6}$ Torr) by an electron gun and depositing it on a heated substrate, in place of the CVD method. In a molecular beam epitaxy MBE method which vapor-deposits in high vacuum ($< 10^{-9}$ Torr), Si beam and SiO_2 start reaction at a substrate temperature of higher than 900°C and the nuclearization of Si on the SiO_2 is stopped. (Journal of Applied Physics, 53, 10, p 6839, 1983, by T. Yonehara, S. Yoshioka and S. Miyazawa).

By using this phenomenon, the single Si nuclear is selectively formed on the fine silicon nitride area on the SiO_2 and the single crystal Si is grown thereon. The deposition condition is vacuum of 10^{-8} Torr, an Si beam intensity of 9.7×10^{14} atoms/cm^2•sec and a substrate temperature of 900 - 1000°C.

A reaction product SiO_2 having a very high vapor pressure is produced by a reaction of $SiO_2 + Si \rightarrow 2SiO \uparrow$ and etching of SiO_2 by Si occurs by the evaporation.

On the other hand, no such etching occurs on the silicon nitride but the nuclearization and deposition occur.

As a material for forming the deposition surface (S_{NDL}) having the high nuclearization density, tantalum oxide (Ta_2O_5) or silicon nitride oxide (SiON) may be used in place of silicon nitride attain the same effect. This material is formed finely as the deposition surface (S_{NDL}) and the single crystal may be grown thereon in the same manner.

The single crystal area is formed on the light transmissive substrate by the single crystal formation method described above.

The photoelectric converter of the present embodiment offers the following advantages.

(1) Since the light transmissive substrate is used, the incident light to the back side can be used, and the aperture area is equal to the cell area. Accordingly, the photoelectric conversion efficiency is improved, the output is increased and the high integration density is attained.

(2) By machining the light transmissive substrate into the lens, the light applied to the device isolation area can be directed into the photoelectric conversion cell. Thus, a light collection efficiency is improved and a cross-talk is reduced.

(3) By forming the anti-reflection coating or color filter on the light transmissive substrate, the substrate may be used as a protective film to protect from the contamination material which has been a defect in a conventional on-chip filter.

(4) By covering the entire single crystal area by the polysilicon film 19 to block the light, the affect by the light irradiation can be substantially eliminated and a reference output is produced in a dark state.

Claims

1. A photoelectric converter comprising a semiconductor transistor having a main electrode region of one conductivity type semiconductor and a control electrode region of an opposite conductivity type semiconductor, and a capacitor for controlling a potential of the floating control electrode region, wherein carriers generated by a light are stored in said control electrode region by controlling the potential of said control electrode area through said capacitor, a stored voltage is read and stored carriers are refreshed,

characterized by that, on an unnucleation surface (S_{NDS}) of said light transmissive substrate, a nucleation surface (S_{NDL}) of a heterogeneous material having a sufficiently higher nucleation density than that of the material of unnucleation surface and a sufficiently small area to permit growth of single unclear of the deposit material is formed, a single crystal region is grown around the single nuclear formed in the nucleation surface (S_{NDL}), and a

photoelectric conversion cell is formed in the single crystal region.

2. A photoelectric converter according to Claim 1 wherein the other side of said light transmissive substrate having nucleation surface is machined to collect light into the single crystal region.

3. A photoelectric converter according to Claim 1 wherein a single-layer or multi-layer light transmissive layer having a different optical property is formed on the other side of said light transmissive substrate having the nucleation surface.

4. A photoelectric converter according to Claim 1 wherein at least one opaque layer is formed in a recess formed in said light transmissive substrate.

5. A method of making a semiconductor device comprising the steps of providing on a substrate in a predetermined region thereof a nucleation surface adapted and arranged to permit the formation of only a single nucleus in the region, forming the single nucleus, and growing a single crystal in the region around the nucleus.

6. A semiconductor device made by the method of claim 5.

Fig. 1

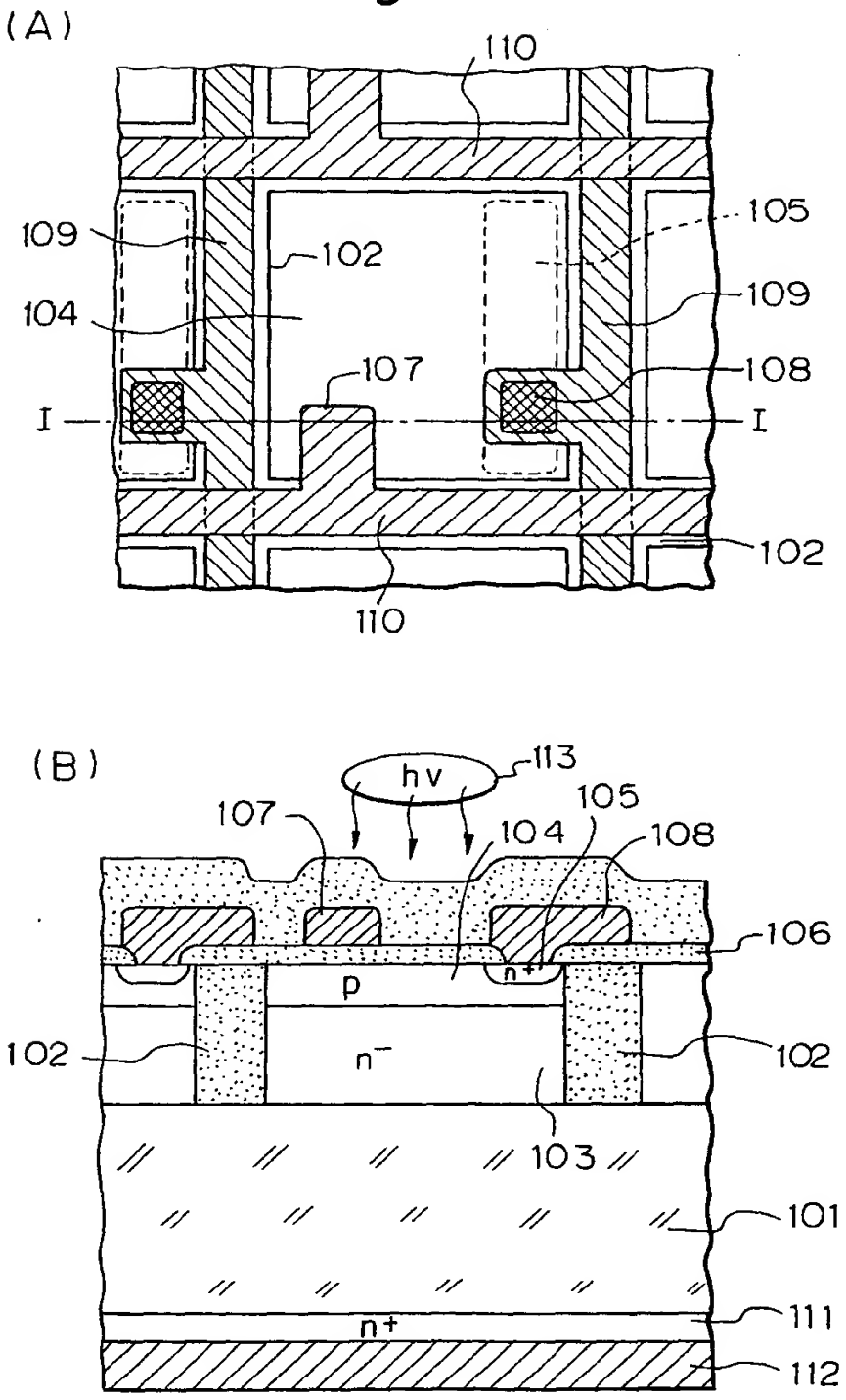


Fig. 2

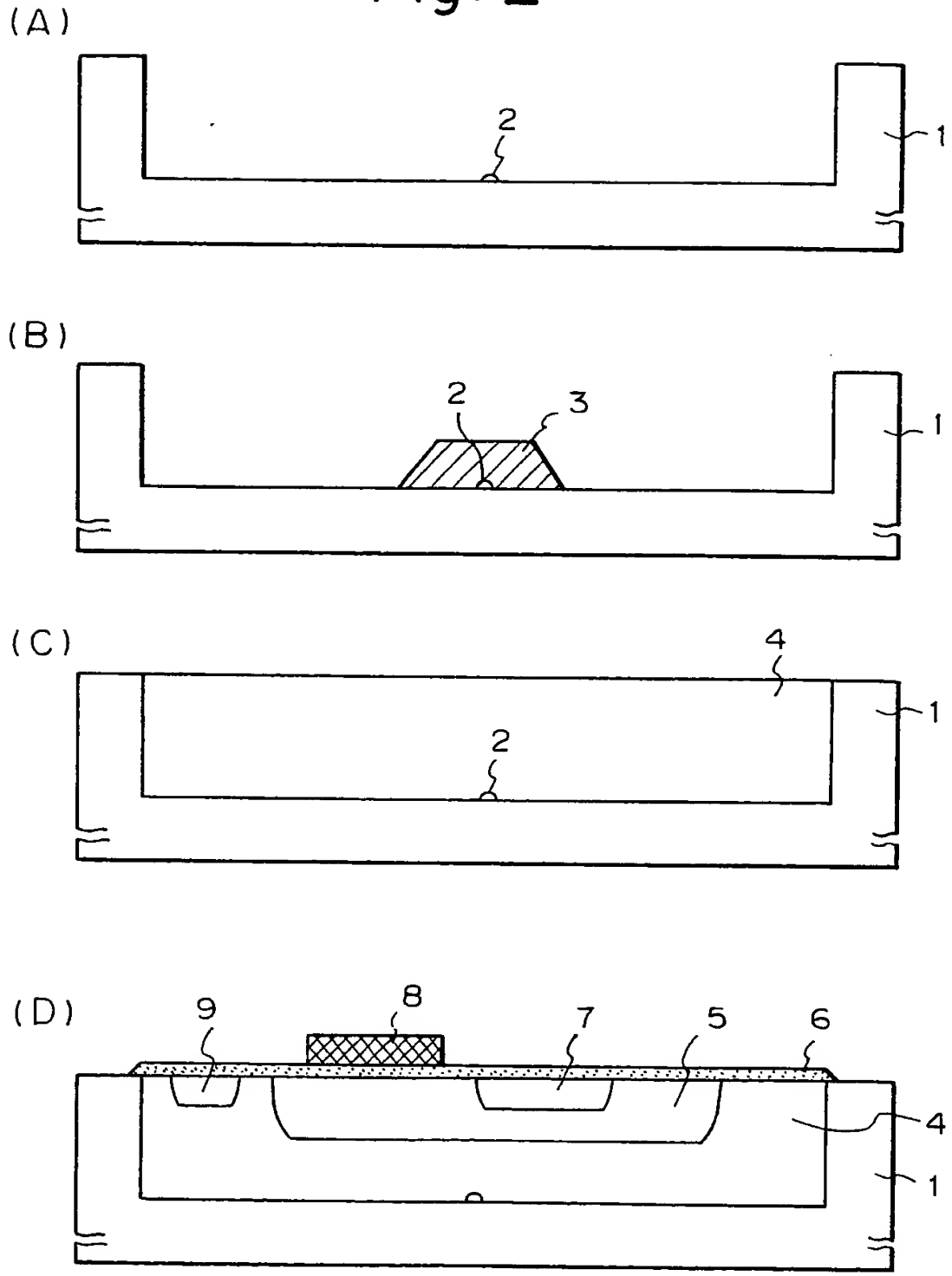


Fig. 2

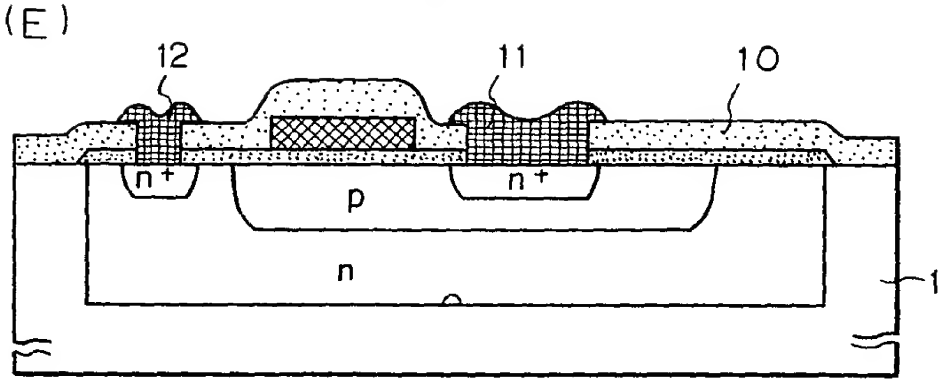


Fig. 3

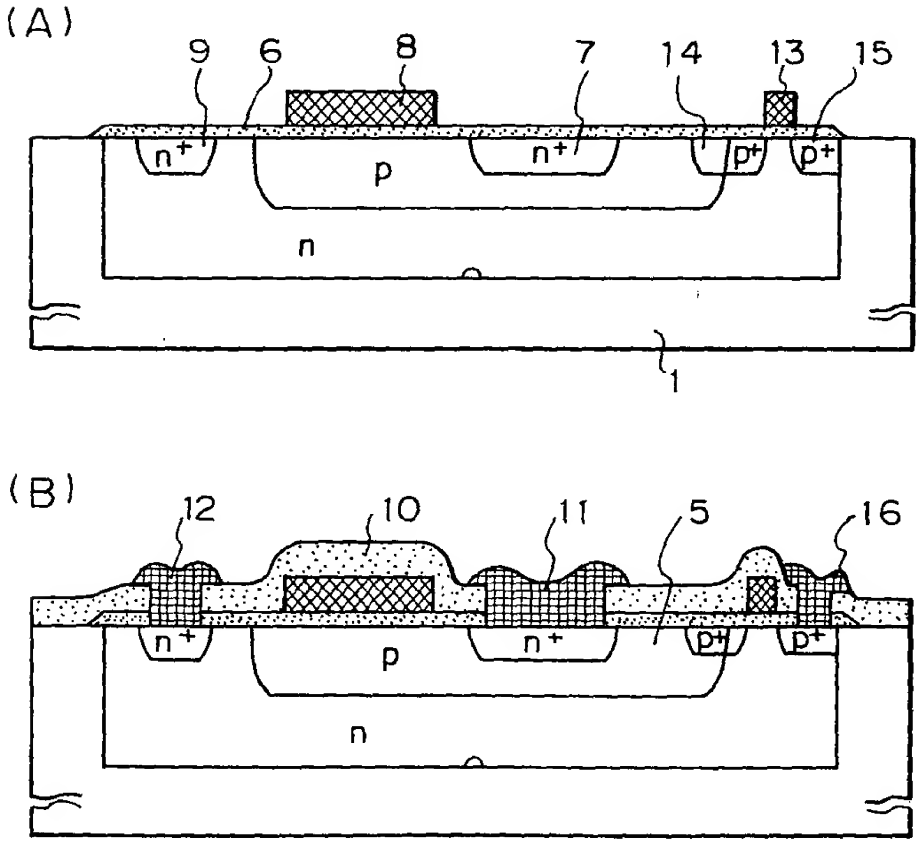


Fig. 4

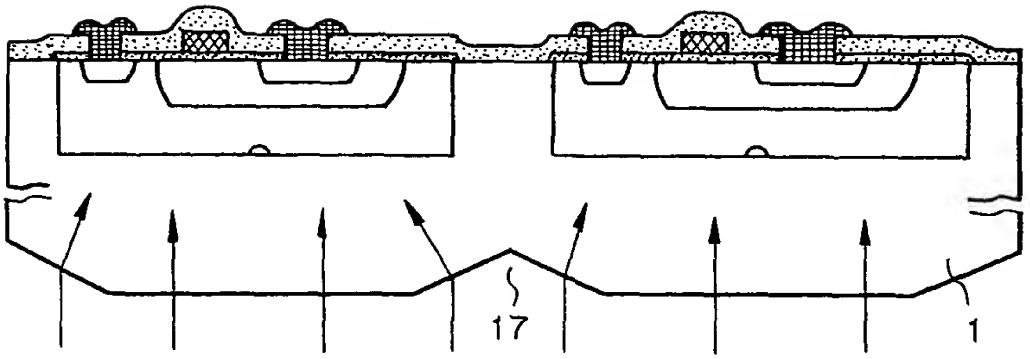


Fig. 5

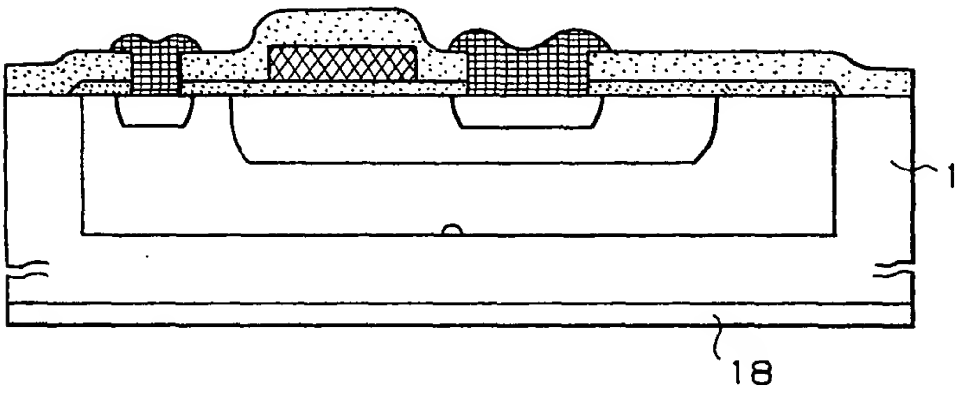


Fig. 6

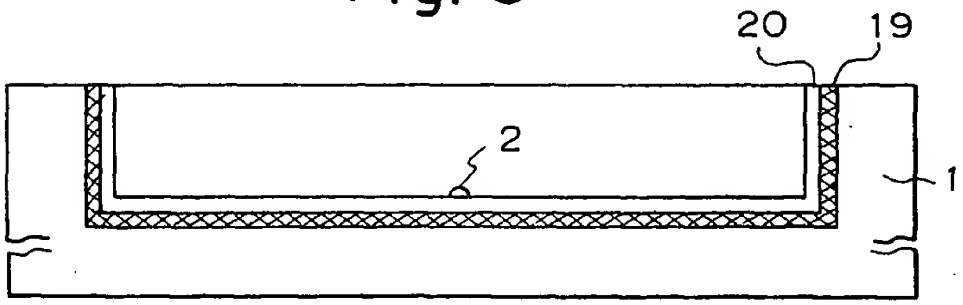


Fig. 7

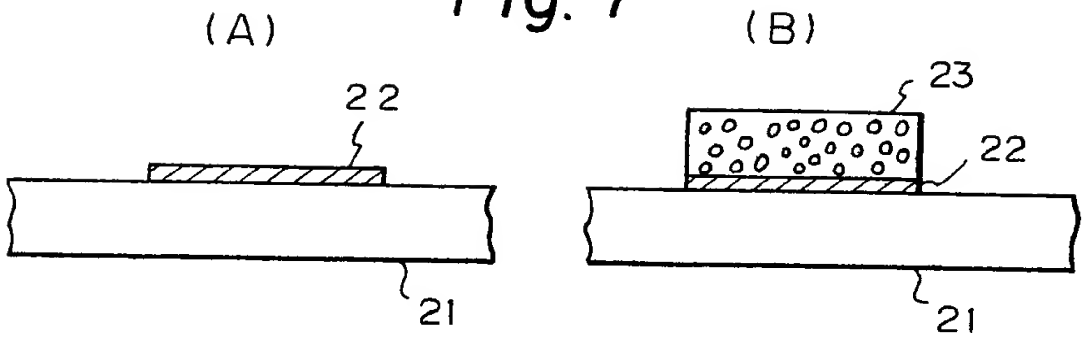
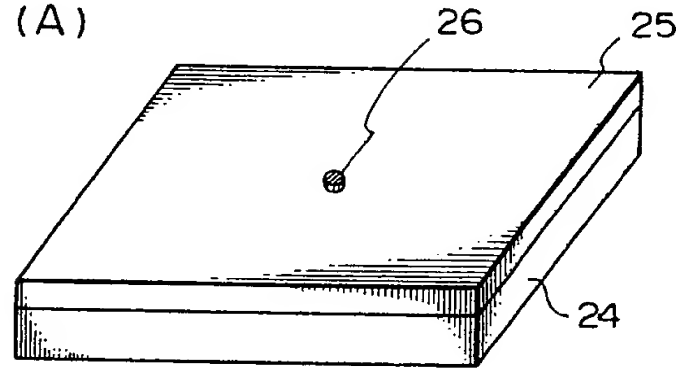


Fig. 10



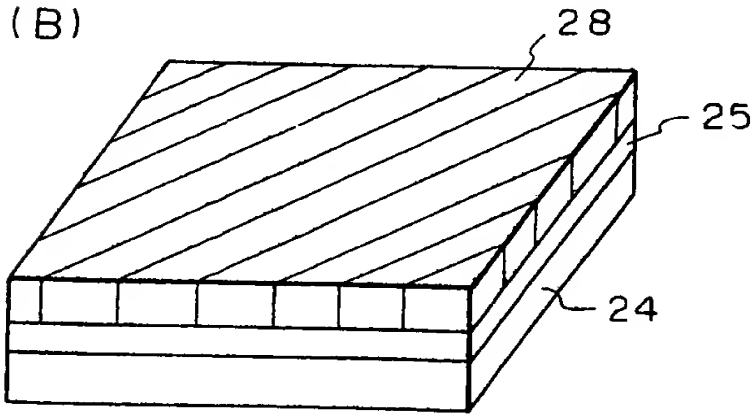


Fig. 8

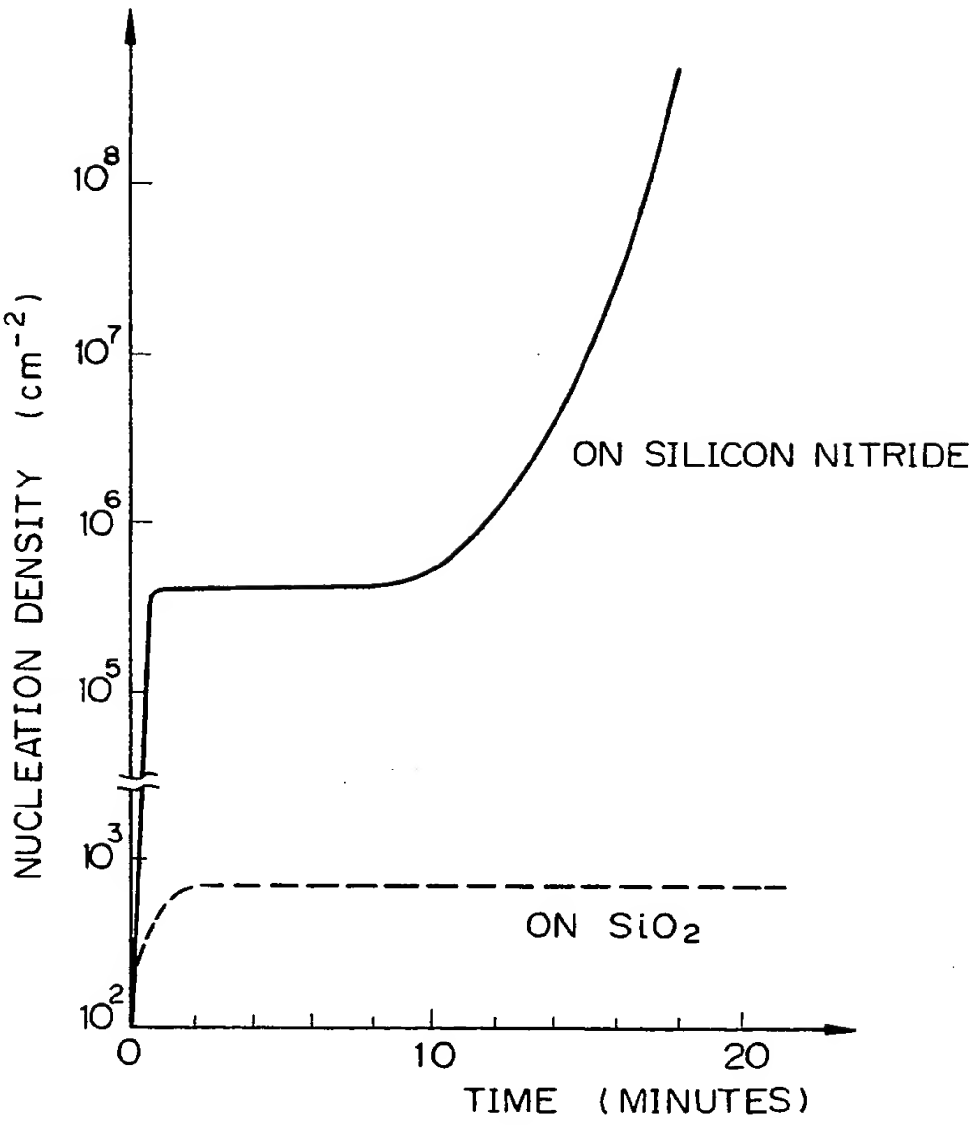


Fig. 9

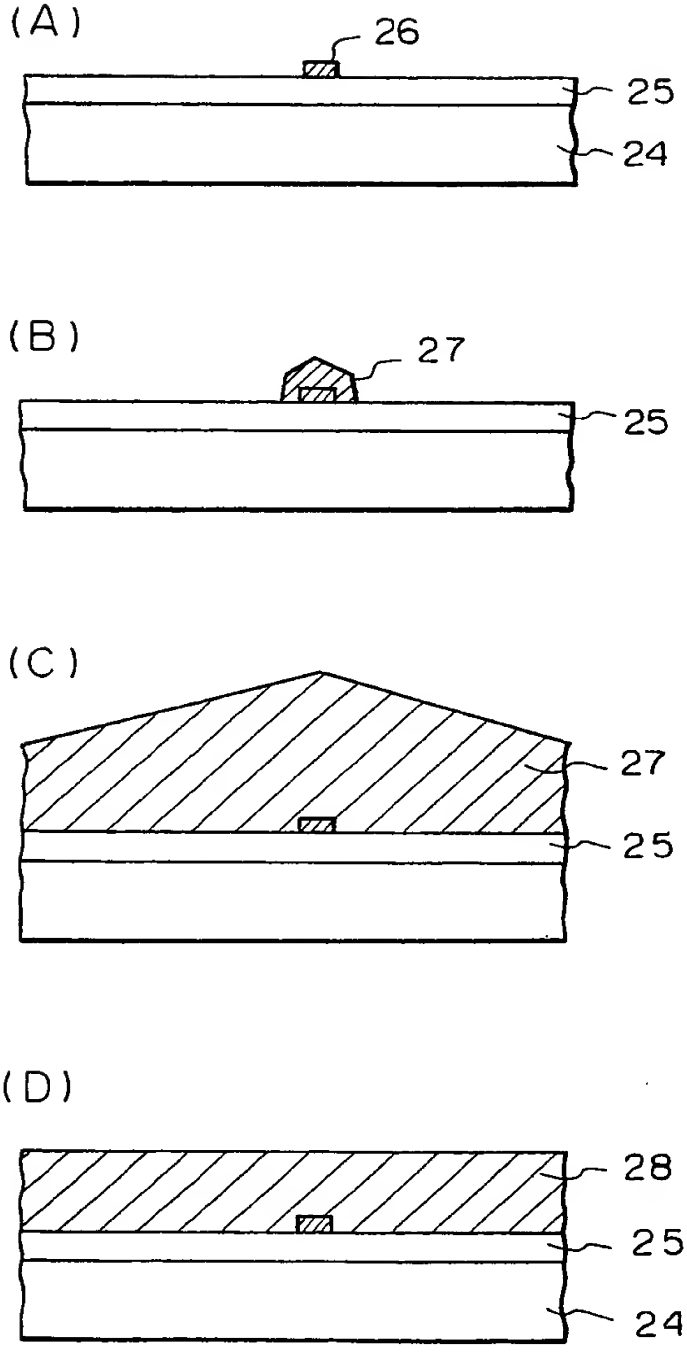


Fig. 11

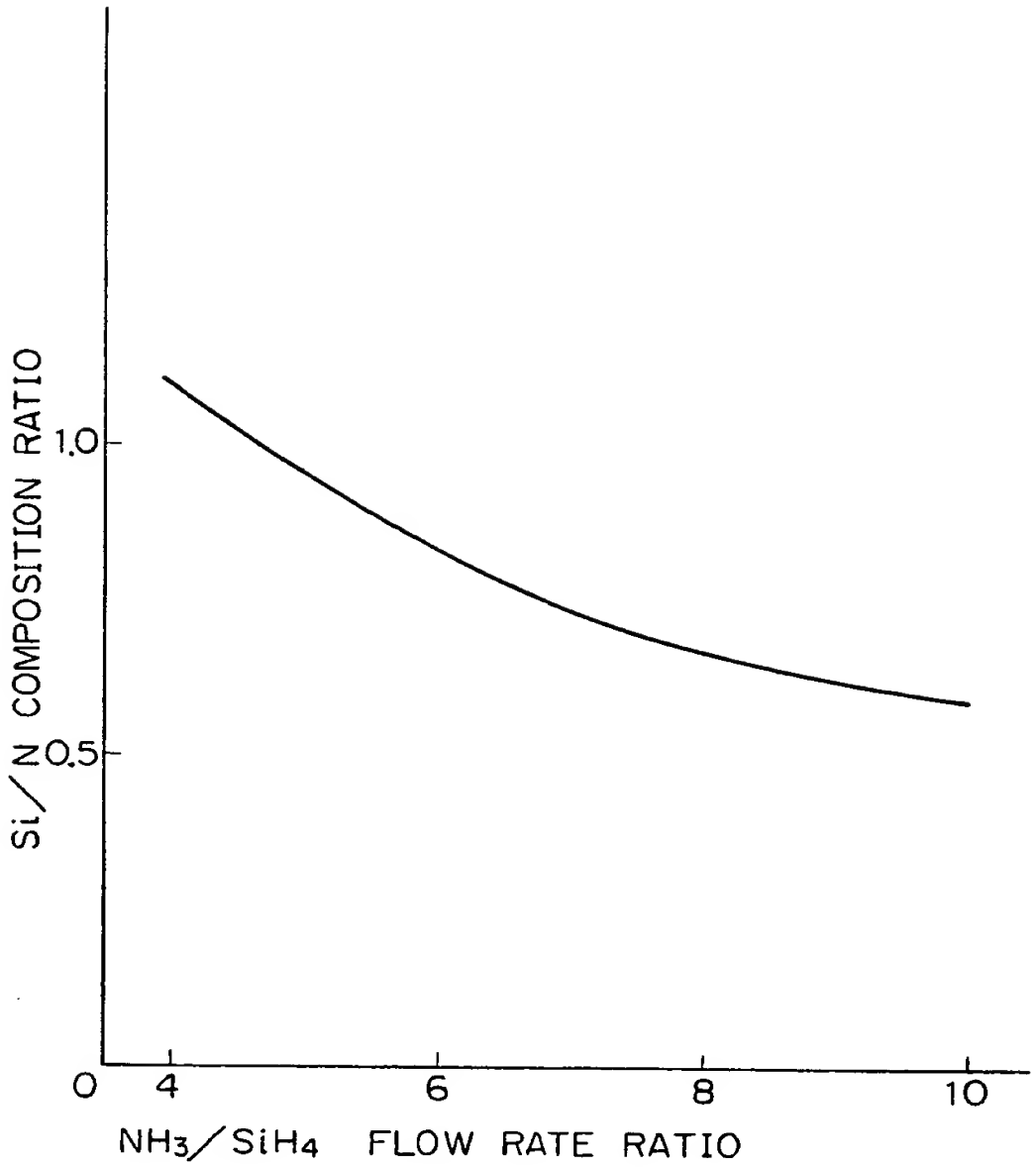


Fig. 12

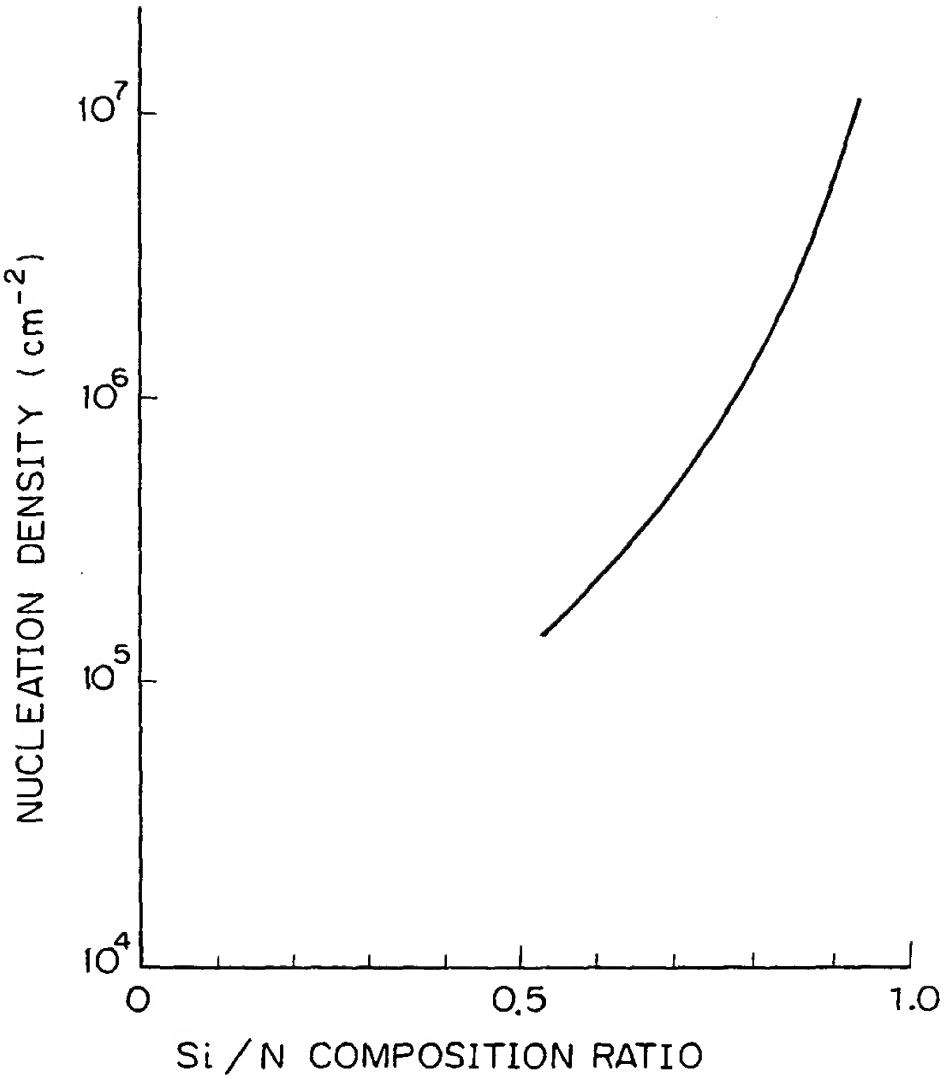
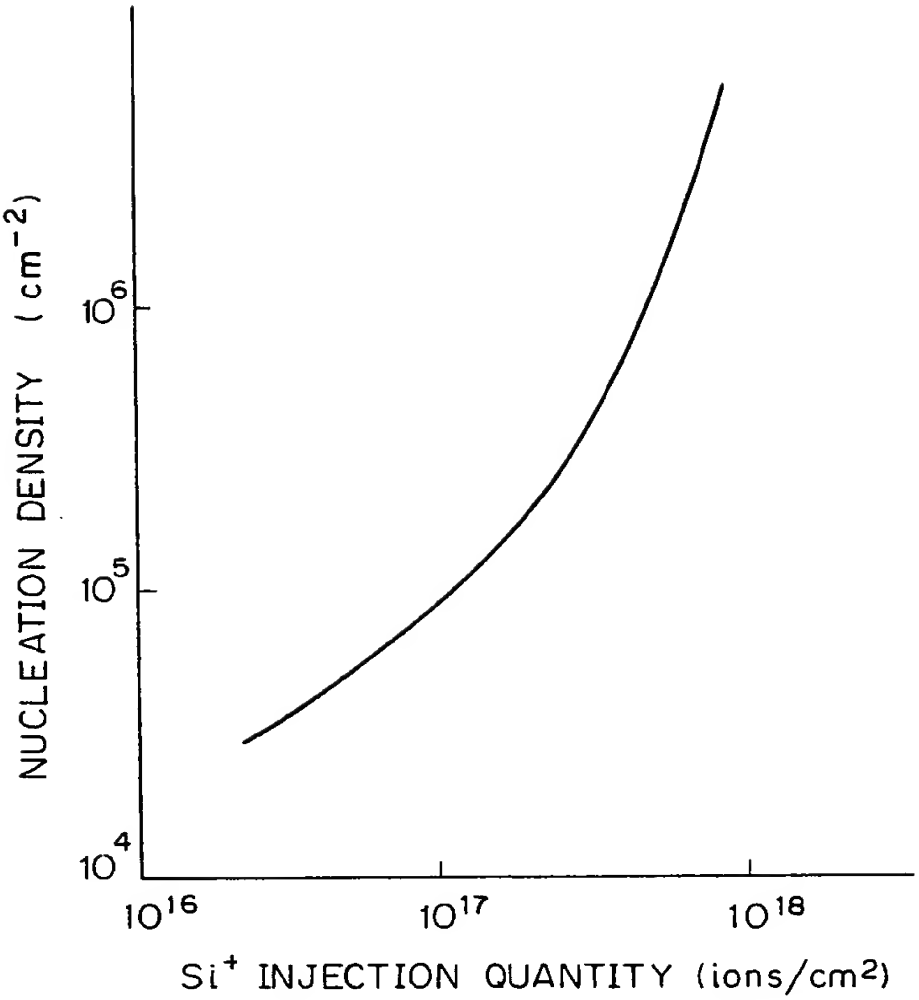


Fig. 13



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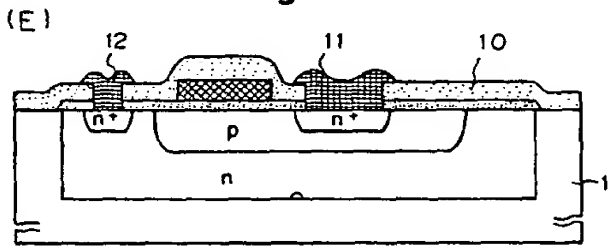
(54) **Photoelectric converter.**

(57) A photoelectric converter comprises a semiconductor transistor having a main electrode region of one conductivity type semiconductor and a control electrode region of an opposite conductivity type semiconductor, and a capacitor for controlling a potential of the floating control electorde region. Carriers generated by a light are stored in the control electrode region by controlling the potential of the control electrode area through the capacitor. The stored voltage is read and stored carriers are refreshed.

On an unnucleation surface (S_{NDS}) of said light transmissive substrate, a nucleation surface (S_{NDL}) of a heterogeneous material having a sufficiently higher nucleation density than that of the material of unnucleation surface and a sufficiently small area to permit growth of single nuclear of the deposite surface is formed. A single crystal region is grown around the single nuclear formed in the nucleation surface (S_{NDL}). And a photoelectric conversion cell is formed is the single crystal region.

Fig. 2



EP 0 260 955 A3

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EUROPEAN SEARCH REPORT

Application Number

EP 87 30 8200

DOCUMENTS CONSIDERED TO BE RELEVANT

Category	Citation of document with indication, where appropriate, of relevant passages	Relevant to claim	CLASSIFICATION OF THE APPLICATION (Int. Cl.4)
Y,D	EP-A-0 132 076 (OHMI) * Page 15, line 1 - page 18, line 19; claims; page 72, line 5 - page 76, line 21; figures 1,16 *	1,5,6	H 01 L 31/10 H 01 L 31/18
Y	JOURNAL OF THE ELECTROCHEMICAL SOCIETY, vol. 127, no. 1, January 1980, pages 194-202, Manchester, US; W.A.P. CLAASSEN et al.: "The Nucleation of CVD silicon on SiO2 and Si3N4 substrates 1. The SiH4-HCI-H2 system at high temperatures" * Whole article *	1,5,6	
A	JOURNAL OF THE ELECTROCHEMICAL SOCIETY, vol. 130, no. 7, July 1983, pages 1571-1580, Manchester, New Hampshire, US; L. JASTRZEBSKI et al.:"Growth process of silicon over SiO2 by CV: Epitaxial Lateral Overgrowth Technique" * Whole article *	1,5,6	
A	EP-A-0 154 962 (HITACHI LTD) * Page 6, line 17 - page 7, line 26; figure 5A *	1,4	TECHNICAL FIELDS SEARCHED (Int. Cl.4) H 01 L
A	PATENT ABSTRACTS OF JAPAN, vol. 11, no. 33 (E-476)[2480], 30th January 1987; & JP-A-61 199 661 (TOKYO ELECTRIC CO., LTD) 04-09-1986 * Abstract *	1,4	
A	PATENT ABSTRACTS OF JAPAN, vol. 9, no. 302 (E-362)[2025], 29th November 1985; & JP-A-60 140 879 (FUJITSU K.K.) 25-07-1985 * Abstract *	1,2	

The present search report has been drawn up for all claims

Place of search	Date of completion of the search	Examiner
THE HAGUE	09-03-1989	LINA F.

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